Observation of the Superconducting Proximity Effect in the Surface State of SmB₆ Thin Films

Seunghun Lee, ^{1,2} Xiaohang Zhang, ^{1,2} Yangang Liang, ^{1,2} Sean W. Fackler, ^{1,2} Jie Yong, ² Xiangfeng Wang, ² Johnpierre Paglione, ² Richard L. Greene, ² and Ichiro Takeuchi ^{1,2,*}

¹Department of Materials Science and Engineering, University of Maryland,
College Park, Maryland 20742, USA

²Center for Nanophysics and Advanced Materials, Department of Physics, University of Maryland, College Park, Maryland 20742, USA

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The proximity effect at the interface between a topological insulator and a superconductor is predicted to give rise to chiral topological superconductivity and Majorana fermion excitations. In most topological insulators studied to date, however, the conducting bulk states have overwhelmed the transport properties and precluded the investigation of the interplay of the topological surface state and Cooper pairs. Here, we demonstrate the superconducting proximity effect in the surface state of SmB₆ thin films which display bulk insulation at low temperatures. The Fermi velocity in the surface state deduced from the proximity effect is found to be as large as 10⁵ m/s, in good agreement with the value obtained from a separate transport measurement. We show that high transparency between the topological insulator and a superconductor is crucial for the proximity effect. The finding here opens the door to investigation of exotic quantum phenomena using all-thin-film multilayers with high-transparency interfaces.

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I. INTRODUCTION

Topological insulators (TI), which show exotic metallic surfaces with insulating bulk, have attracted tremendous interest in the condensed matter physics community. The topologically protected surface state of TIs has been regarded as a promising platform for exploring exotic quantum phenomena in solid materials. In particular, it has been suggested that the superconducting proximity effect occurring at an interface between a TI and a superconductor may give rise to chiral topological superconductivity [1–3] and Majorana fermion excitations [4–9]. To this end, concerted experimental efforts have been made to study the superconducting proximity effect in superconductor-TI bilayer structures. However, in most Bi- and Te-based Tis, such as Bi₂Se₃, Bi₂Te₃, Sb₂Te, etc., the overwhelming conducting bulk electronic states hinder interfacial quantum phenomena and preclude the investigation of the interplay of the topological surface state and Cooper pairs [10–13]. In addition, low transparency at the superconductor-TI interface due to a nonpristine surface of the TI can significantly

Published by the American Physical Society under the terms of the Creative Commons Attribution 3.0 License. Further distribution of this work must maintain attribution to the author(s) and the published article's title, journal citation, and DOI. reduce the extent of the proximity effect [5,14–16]. Therefore, suppressing the bulk conductivity and securing high interfacial transparency in superconductor-TI bilayers have been an outstanding issue in this field.

Samarium hexaboride (SmB₆) has recently emerged as an ideal material to explore TI-based quantum phenomena because of its true insulating bulk [17]. To date, many theoretical and experimental studies have provided strong evidence for the existence of the conducting surface state and a robust insulating bulk of SmB₆ at low temperatures [18–38]. Specifically, electronic transport measurements on SmB₆ single crystals have revealed surface-current-dominated conduction [20,31] and thickness-independent Hall voltage at low temperatures due to the conducting surface [20]. A point contact spectroscopy study has indicated the presence of an insulating bulk in SmB₆ at low temperatures [23,39].

Even though there still remain questions regarding the nature of the surface state of SmB₆, including the origin of the metallic surface state and whether the surface state is truly topologically protected [17,22,27,40–42], recent angle-resolved photoemission spectroscopy (ARPES) and quantum oscillation experiments have provided strong evidence in support of the predicted nontrivial topological nature of the surface state in SmB₆ [19,32–34]. In particular, a circular dichroism APRES measurement has demonstrated the spin-polarized surface state of SmB₆ as a signature of a topologically protected surface state

^{*}Corresponding author. takeuchi@umd.edu

[32,34]. We have previously demonstrated successful fabrication of SmB_6 thin films using a cosputtering method [37]. The electronic transport properties of the SmB_6 thin films were found to be consistent with those of single crystalline SmB_6 , and they support the topological Kondo insulator picture of SmB_6 [36,37].

In this paper, we demonstrate direct observation of the proximity effect induced in the surface state of SmB₆ for the first time. We apply the Usadel formalism to the systematic change in the superconducting transition temperature of a series of *in situ* deposited SmB₆/Nb bilayers with different Nb layer thicknesses and arrive at the normal coherence length of the surface state of SmB₆ to be \approx 9 nm. From a separate transport study on SmB₆ thin films, we find the thickness of the surface state to be \approx 6 nm. The Fermi velocity (v_F) obtained from the transport measurements is in good agreement with v_F calculated using the normal coherence length, and it is of the order of 10^5 m/s. These numbers paint a picture of a material whose bulk insulation provides a unique opportunity to probe and exploit its surface state in thin-film multilayer devices.

II. METHODS

SmB₆ thin films are prepared by cosputtering of SmB₆ and B targets; additional B sputtering is performed for compensating B deficiency, and the base pressure is $\sim 2 \times 10^{-8}$ Torr. The thin films are grown on a Si (100) substrate at 800 °C in an atmosphere of Ar (99.999%) and the working pressure is kept at 10 mTorr. After deposition, further annealing is performed at 800 °C for 3 h in high-vacuum. X-ray diffraction and transmission electron microscopy results can be found in Ref. [37]. For fabricating Nb/SmB₆ bilayers, Nb layers are prepared on SmB₆ by the sputtering method both *in situ* and *ex situ* at room temperature. The base pressure is $\sim 2 \times 10^{-8}$ Torr and the working pressure is kept at 7 mTorr. For fabricating the Nb/Au bilayer, a Au layer is prepared on the Si substrate by a thermal evaporator and the base pressure is $\sim 10^{-6}$ Torr.

The electrical properties are measured by a physical properties measurement system. The temperature-dependent resistance for T_c evaluation is measured by a simple four-point probe method with 0.01 K step. Electrical contacts are made with an Al wire bonder, and the typical contact resistance is $\sim 1\,\Omega$. Transport characteristics of SmB₆ thin films are measured with Hall bar geometry; six-contact 1-2-2-1 Hall bars with 200- μ m channel size are prepared by an ion milling process.

III. RESULTS AND DISCUSSION

A. Transport characteristics

We perform transport measurements on SmB_6 thin films with varying film thickness. In order to perform the transport measurements, each film is patterned into a Hall bar with a channel width of 200 μ m. Resistance

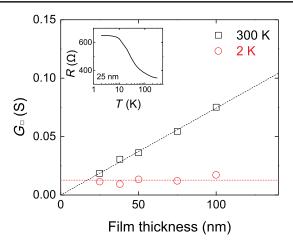


FIG. 1. Sheet conductance (G_{\square}) of SmB₆ thin films at 2 and 300 K as a function of film thickness. Dashed lines are linear fits to the experimental results. The slope of 300 K data represents 3D bulk conductivity at 300 K in SmB₆ thin films, and the fit to the data (black dot line) extrapolates to zero at zero thickness, consistent with the fact that the conduction in SmB₆ thin films is dominated by a 3D bulk transport at 300 K. At 2 K, G_{\square} is independent of the thickness, and the guide line (red dotted line) is the fit to the data, which indicates the average value of G_{\square} at 2 K. The inset shows resistance versus temperature curve for a 25-nm SmB₆ thin film.

versus temperature (R-T) is measured for all SmB₆ films, and their sheet conductance (G_{\square}) values at 300 and 2 K are plotted in Fig. 1. G_{\square} is calculated using $G_{\square} = G \cdot a/w$, where a and w are the length and the width of the Hall bar channels, respectively. All samples show a nearly constant resistance value at low temperatures. The inset shows a representative R-T curve for a 25-nm thin film. The result is consistent with previous reports [24,29,31,35] where a saturation of resistance (i.e., plateau) has been attributed to the transition from bulk-state-dominated conduction to surface-state-dominated conduction at low temperatures.

To see if there is any thickness-dependent effect in our samples, we plot G_{\square} at 2 and 300 K as a function of film thickness (Fig. 1). G_{\square} at 300 K clearly displays a linear thickness dependence, indicating bulk transport. However, if the resistance plateau at low temperatures arises from the surface conduction, G_{\square} at 2 K should be independent of the thickness and become a constant. This is indeed the case, as shown in Fig. 1, strongly suggestive of the presence of the surface conduction channel. We also measure the Hall resistance (R_{xy}) as a function of magnetic field, and calculate the sheet carrier concentration and the mobility of the SmB₆ thin films at 2 K. Both are found to be constant and independent of the film thickness (see Supplemental Material [43]).

To determine the thickness of the surface conduction channel, we adopt a simple parallel conductance model where the electronic conduction is through two channels: the surface channel including contributions from both the top and the bottom surfaces and the bulk channel [24,31,44]. This model is valid for the temperature region where the Kondo gap is fully open, which is a sufficient condition for the presence of the surface state. Thus, we apply this model to the transport results in the temperature range below 30 K, where the Kondo hybridization gap has been reported to fully open in SmB_6 [23,25,45,46]. The conductance of the surface channel is assumed to be temperature independent, and that of the bulk channel is temperature dependent and it can be described by a bulk resistivity and exponential function (i.e., Arrhenius equation). The total conductance of the SmB_6 thin film can then be described as

$$G = G_{\text{surface}} + G_{\text{bulk}},\tag{1}$$

$$G_{\text{surface}} = \frac{w}{a} G_{\square, 2 \text{ K}},$$
 (2)

$$G_{\text{bulk}} = \frac{w \cdot t_{\text{bulk}}}{a} \cdot \sigma_{\text{bulk},300 \text{ K}} \exp\left(-\frac{E_a}{k_B T} + \frac{E_a}{300 k_B}\right). \tag{3}$$

The use of the bulk conductivity in the G_{bulk} term allows us to add a thickness of bulk channel (t_{bulk}) as fitting parameters, and subsequently to estimate a thickness of surface channel (t_{surface}) because the total film thickness is $2t_{\text{surface}} + t_{\text{bulk}}$. The bulk conductivity $\sigma_{\text{bulk},300 \text{ K}}$ is $7.4 \times 10^5 \text{ S m}^{-1}$, which is the slope of G_{\square} at 300 K obtained from the data shown in Fig. 1. $G_{\square,2 \text{ K}}$ is the G_{\square} at 2 K for each sample, and a and w are the length and the width of the Hall bar channel, respectively. Fitting parameters are t_{bulk} and E_a , where E_a is an activation energy. We fit the data for all samples using Eqs. (1)–(3). A representative fitting result is shown in Fig. 2(a). The values of fitting parameters E_a and t_{surface} for the different thickness samples are shown in Fig. 2(b), and t_{surface} is the thickness of the surface conduction channel, i.e., the

thickness of the surface state of SmB₆. Both E_a and $t_{\rm surface}$ are found to be constant regardless of the film thickness. The average value of E_a is, 3.1 ± 0.1 meV, which is in good agreement with the values obtained from SmB₆ single crystals [24,28,31].

From the above analysis, we obtain $t_{\text{surface}} \approx 6$ nm. To obtain v_F , we use $v_F = E_g t_{\text{surface}}/h$ derived from the solution of the effective low-energy Hamlitonian for topological insulators with a surface state where E_q is the bulk energy gap [29,47]. We note that E_a obtained from transport measurements does not correspond to the intrinsic hybridization gap of SmB₆, because the activation energy related to the transport characteristics of the bulk state of SmB₆ may be affected by the presence of residual states in the gap or other activation behaviors [39,46]. Instead, we use E_q of 17 meV measured on our SmB₆ thin films by terahertz spectroscopy [48], which is in good agreement with E_q values obtained from single crystal SmB₆ using point contact spectroscopy [23], scanning tunneling spectroscopy [25], etc. With $E_g = 17$ meV and $t_{\text{surface}} = 6$ nm, we arrive at $v_F \sim 1.6 \times 10^5$ m/s, a value close to that measured in quantum oscillation measurements on single crystal SmB₆ [19]. This value is also in agreement with an independent estimate of v_F obtained from the proximity effect study as discussed below.

B. Proximity effect

The superconducting proximity effect describes a phenomenon at a superconductor-normal metal interface where Cooper pairs diffuse into the normal metal resulting in the suppression of the critical temperature (T_c) of the superconductor while inducing surface or local superconductivity in the normal metal. We fabricate Nb/SmB₆ bilayers and observe a change in T_c depending on the thickness of the Nb layer due to the proximity effect at the interface. To characterize the proximity effect of the bilayers, we treat

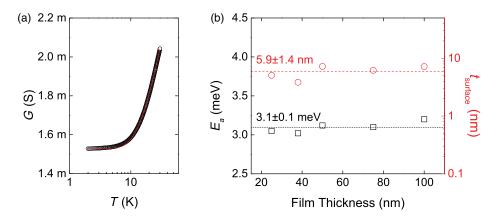


FIG. 2. Estimating the thickness of the surface conduction channel in SmB_6 thin films. (a) Conductance versus temperature of a 25-nm SmB_6 thin film. The red line is a best fit to the experimental data using the parallel conductance model. (b) Activation energy (E_a) and the thickness of the surface conduction channel $(t_{surface})$ extracted from the fit as a function of the film thickness for SmB_6 .

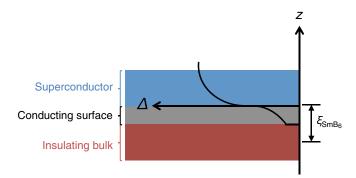


FIG. 3. Model for the proximity effect for superconducting-TI system. Given the surface conducting channel of the SmB₆ as discussed in the text, we adopt the model of the proximity effect for the superconducting layer and the surface conduction channel, the surface state of SmB₆. The plot in the schematic represents the evolution of the superconducting pair potential $[\Delta(z)]$. In this model, we consider the case where the normal coherence length of SmB₆ ($\xi_{\rm SmB_6}$) is larger than the thickness of the surface state of SmB₆ ($t_{\rm surface}$).

the Nb/SmB₆ bilayer as a superconductor-metal bilayer system, where the metallic layer in SmB₆ is the surface conducting channel (Fig. 3), and we calculate the normal coherence length and v_F of SmB₆ thin films using the Usadel equation [16,49,50]. Assuming that the normal coherence length of SmB₆ ($\xi_{\rm SmB_6}$) is longer than the thickness of the surface conducting channel of SmB₆ ($\xi_{\rm SmB_6} > t_{\rm surface}$), the fitting equation for evaluating $\xi_{\rm SmB_6}$ can be obtained by linearizing the Usadel equation [16]:

$$\frac{T_{cb}}{T_{cs}} = 1 - \frac{\pi^2}{4} \frac{\xi_{\text{Nb}}}{d_{\text{Nb}}} \gamma \cdot p(\gamma), \tag{4}$$

$$\gamma = \frac{\rho_{\text{Nb}}\xi_{\text{Nb}}t_{\text{surface}}}{\rho_{\text{SmB}_6}(\xi_{\text{SmB}_6})^2},\tag{5}$$

$$p(\gamma) = 1.17 + \frac{2}{\pi^2} \ln(1 + 0.98\gamma^{-2}),$$
 (6)

where T_{cb} and T_{cs} represent the T_c of the Nb/SmB₆ bilayer and the T_c of a single Nb layer, respectively. T_{cb} is evaluated by the Ginzburg-Landau equation, and γ and $p(\gamma)$ are approximated by the expressions above [16]. γ represents the strength of the proximity effect between layers [49]. $\rho_{\rm Nb}$ and $\rho_{\rm SmB_6}$ are the residual resistivities of the Nb layer and the SmB₆ layer, respectively. $\xi_{\rm Nb}$ is the superconducting coherence length of the Nb layer. $d_{\rm Nb}$ is the thickness of the Nb layer. To check the validity of the model, a series of Nb/Au bilayers are also fabricated, and the extracted value of normal coherence length of Au ($\xi_{\rm Au}$) is found to be in the range of its previously reported values [51,52], and it is also consistent with the clean limit value of $\xi_{\rm Au}$.

For Nb/SmB₆ bilayers, $d_{\rm Nb}$ is varied from 20 to 100 nm, and different thicknesses of SmB₆ layers (10, 25, and

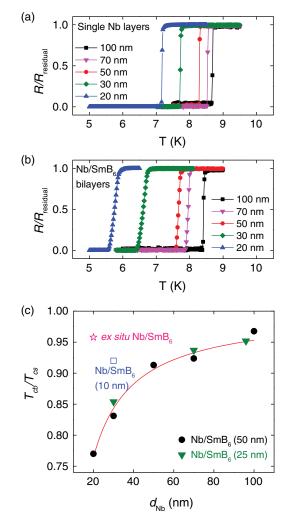


FIG. 4. Proximity effect in Nb/SmB₆ bilayer. Resistance versus temperature curves of (a) single Nb layers and (b) Nb/SmB₆ bilayers for different Nb layer thicknesses ($d_{\rm Nb}$) where the SmB₆ layer thickness is fixed to be 50 nm. Each Nb layer is deposited on SmB₆ in situ. The resistance values are normalized by values obtained at temperatures slightly above each transition temperature ($R_{\rm residual}$). (c) T_{cb}/T_{cs} (i.e., ratio of T_c of Nb/SmB₆ bilayer to T_c of single Nb layer) as a function of $d_{\rm Nb}$. The data points from bilayers with different SmB₆ layer thicknesses (10 and 25 nm) are also included. The solid line is a fit using the Usadel equation. The star indicates T_{cb}/T_{cs} of the Nb/SmB₆ bilayer where 20-nm Nb is deposited on SmB₆ (50 nm) after its surface is first exposed to air, forming an ex situ interface.

50 nm) are used. Figures 4(a) and 4(b) show the resistance versus temperature curves of single Nb layers and the bilayers with 50-nm-thick ${\rm SmB_6}$ near T_c for different Nb layer thicknesses ($d_{\rm Nb}$), respectively. The resistance values are normalized by values obtained at temperatures slightly above each transition temperature (${\rm R_{residual}}$). The T_c of a single Nb layer gradually decreases with decreased $d_{\rm Nb}$ due to localization [53], and each Nb/SmB₆ bilayer has a lower T_c than the corresponding Nb single layer due to the proximity effect established between Nb and SmB₆ layers.

From x-ray diffraction and cross-sectional SEM of the bilayers, no signature of diffusion is detected at the interface. We measure the roughness of the SmB₆ layers, and they are found to be very smooth (with a rms roughness of 1.4 nm for a 5 μ m \times 5 μ m area). Thus, we believe that the roughness of the film is not responsible for the observed decrease of T_c .

Figure 4(c) shows the T_{cb}/T_{cs} of the Nb/SmB₆ bilayers as a function of $d_{\rm Nb}$. As a comparison, one bilayer is made with an *ex situ* interface: following the deposition of SmB₆, it is exposed to air before the Nb layer is deposited. This *ex situ* sample shows a much higher T_{cb}/T_{cs} (i.e., higher T_c) than the corresponding *in situ* bilayer. The proximity effect is very sensitive to the nature of the interface, and degradation of the topmost surface layer of SmB₆ (e.g., due to oxidation) can reduce the boundary transparency, resulting in reduction of the proximity effect. This result demonstrates that *in situ* formed multilayers with pristine interfaces are critical for establishing the proximity effect at the interface of SmB₆ and a superconductor.

To evaluate ξ_{SmB_6} , we perform a fitting process based on the expression discussed above. Using measured values of $\rho_{\rm Nh} = 8.5 \times 10^{-6} \ \Omega \, {\rm cm}$, the mean-free path (l) of Nb was estimated from $\rho_{\text{Nb}}l = 3.75 \times 10^{-6} \,\mu\Omega \,\text{cm}^2$ [54] (i.e., l = 4.4 nm) and we obtain $\xi_{\text{Nb}} = 0.852(\xi_{\text{Nb(bulk)}}l)^{1/2} =$ 11 nm, where $\xi_{Nb(bulk)}$ is the known value of coherence length of bulk Nb ($\xi_{\text{Nb(bulk)}} = 38 \text{ nm}$) [55]. The thickness of the surface conducting channel of SmB₆ is evaluated above from our transport result ($t_{\text{surface}} = 6 \text{ nm}$), and the resistivity of the surface state is calculated to be $9.4 \times$ $10^{-5} \Omega \,\mathrm{cm} (= \rho_{\mathrm{SmB}_6})$ at 2 K, based on the above measured sheet conductance ($\rho = 2t_{\text{surface}}/G_{\square}$). The only unknown parameter is ξ_{SmB_6} . The fitting result is shown as a red solid line in Fig. 4(c), and we arrive at $\xi_{\rm SmB_6}$ of 8.7 nm. To calculate v_F , $\xi = (v_F \hbar l / 6\pi k_B T)^{1/2}$ is used [56], where l of the SmB₆ thin film was assumed to be the grain size of the SmB₆ film, ≈ 4 nm [37], and we obtain $v_F \approx 10^5$ m/s at 2 K for SmB₆. This v_F is comparable to the value obtained from the transport study above ($v_F = 1.6 \times 10^5 \text{ m/s}$), which implies that the observed superconducting proximity effect can be attributed to the surface state of SmB₆ thin film. As a comparison, we also carry out a fit using the entire bulk of the SmB₆ thin film (even though we show above that the bulk of the film is insulating): in this case, the fit does not provide a v_F value consistent with the v_F value obtained from the transport study. We also note that our bilayer fabrication method in an ultrahigh vacuum process precludes any degradation or contamination as the origin of the metallic surface state.

The proximity effect is expected to persist for bilayers with thinner SmB₆ (than 50 nm discussed above), and we expect it to be suppressed only when the SmB₆ thickness approaches its normal coherence length. As can be seen in Fig. 4(c), the bilayers with 25-nm SmB₆ show the same

trend as the 50-nm ${\rm SmB_6}$ bilayer series. On the other hand, when the ${\rm SmB_6}$ layer is 10 nm, the proximity effect is suppressed, resulting in a much higher T_{cb}/T_{cs} for this data point. These results provide collective evidence that the observed systematic drop in T_c as function of the Nb layer thickness is indeed due to the superconducting proximity effect

The value of v_F and the effective mass of quasiparticles in the surface state of SmB₆ have been the subjects of much debate. Early theoretical calculations had suggested the presence of relatively heavy Dirac quasiparticles with small v_F (~10³ m/s) [26,57] in the topologically protected surface state, while recent APRES and quantum oscillation studies have pointed to light quasiparticles with large v_F of $> 10^4$ m/s[19,32,33]. To resolve the discrepancy, Alexandrov et al. have recently proposed that Kondo breakdown at the surface of SmB₆ could release unquenched moments at the surface, causing the Dirac point to shift down into the valence band and give rise to large v_F values [58]. We examine the Sm valence state in the surface of our SmB₆ thin films using x-ray photoemission spectroscopy at room temperature and find that it is similar to that of SmB₆ single crystals, which is ~ 2.7 [59] (see Supplemental Material [43]). Therefore, we exclude chemical extrinsic effects as the origin of the large v_F value observed here. We believe our high v_F value is also due to the Kondo breakdown effect [58]. We note that t_{surface} of \approx 6 nm found in this study is consistent with the thickness predicted with the Kondo breakdown effect [58]. It has been reported that SmB₆ has three Dirac cones, each with its own slightly different value of v_F [32,60]. The measured v_F in this study is, therefore, expected to be the average v_F of the three Dirac cones.

With the ξ_{SmB_6} obtained from the fitting above, we are able to evaluate γ , which is a measure of the strength of the superconducting proximity effect. The values of γ for in situ and ex situ bilayers are 7.9×10^{-2} and 1.0×10^{-2} , respectively. This implies that the strength of the proximity effect of the in situ sample is roughly 8 times larger than that of the ex situ sample. The strength of the superconducting proximity effect is naturally one of the most important factors in superconductor-normal-superconductor Josephson junctions, and it directly affects the $I_C R_N$ product. It is interesting to note that significantly reduced $I_C R_N$ products, presumably due to lack of the pristine surface of the TI during the fabrication process, have been observed in various studies on chalcogenide-TI-based Josephson junctions [5,6]. Specifically, the observed $I_C R_N$ products in such studies are typically around 20 μ V, which is far below the theoretical value, i.e., $I_C R_N \sim \pi \Delta(0)/2e \approx 4.7 \,\text{mV}$ (for Nb), for junctions in the clean limit [61,62]. Higher $I_C R_N$ product junctions (and, in particular, high I_C junctions) are always desirable for investigating novel quantum phenomena involving coherence of superconductivity, where a signature of such phenomena might appear in the form of small modulation in the critical current. The present study shows that the strength of proximity effect in the bilayer with an $in\,situ$ interface is nearly an order of magnitude higher than that with an $ex\,situ$ interface. We expect that with identical geometric dimensions, junctions fabricated by $in\,situ$ interface may display a substantially larger I_CR_N than junctions fabricated with an $ex\,situ$ interface. Thus, we demonstrate an important prerequisite for attaining a high I_CR_N junction. A nonideal S-N interface, such as one with a thin tunnel barrier (formed due to an $ex\,situ$ process), can result in scattering at the interface leading to decoherence [7,63]. This may be the reason why the proposed signatures of Majorana fermions have not been observed in Josephson junctions to date.

IV. CONCLUSIONS

In summary, thickness-independent behavior in the transport properties is observed in ultrathin SmB_6 films, and the thickness of the surface state is deduced to be $\approx 6\,\mathrm{nm}$. We provide first direct evidence of the superconducting proximity effect in the surface state of SmB_6 through pristine interfaces with *in situ* deposited Nb layers. The Fermi velocity values of the surface state obtained from the transport measurements and the proximity effect are in good agreement with each other. The present work lays the groundwork for fabricating SmB_6 thin-film-based multilayers and devices for investigating quantum phenomena including Majorana fermion excitations.

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